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(11) **EP 0 729 922 A1**

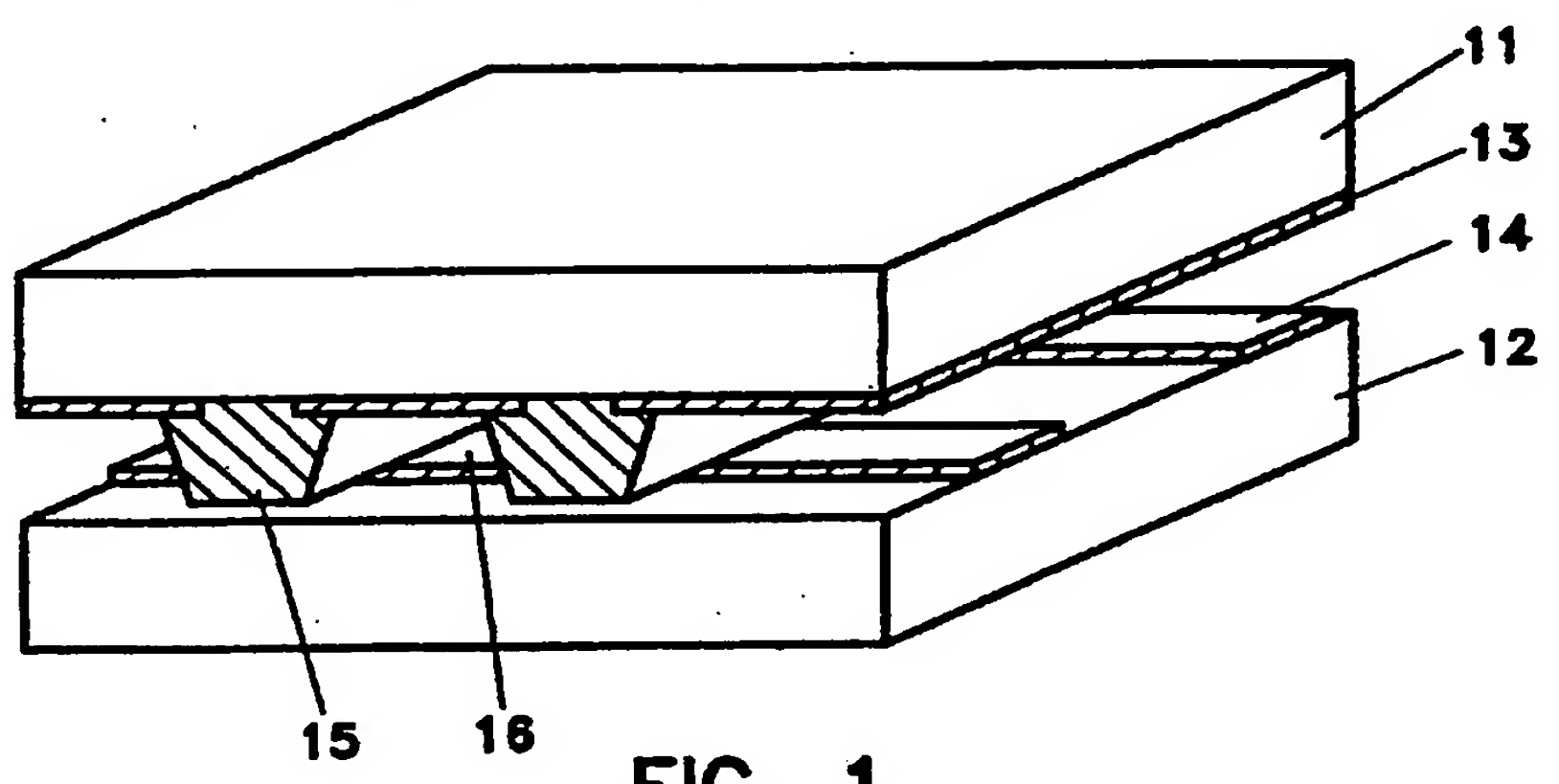
(12) **EUROPEAN PATENT APPLICATION**

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(84) Designated Contracting States: FR NL (30) Priority: 02.03.1995 US 397606 14.02.1996 US 593097 (71) Applicant: CORNING INCORPORATED Corning, N.Y. 14831 (US)	(72) Inventors: • Kohli, Jeffrey Todd Corning, NY 14831 (US) • Moffatt, Dawne Michelle Corning, NY 14831 (US) (74) Representative: Aron, Georges et al Cabinet de Boisse 37, Avenue Franklin D. Roosevelt 75008 Paris (FR)
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(54) **Glasses for display panels**

(57) An aluminosilicate glass having a composition comprising, as calculated in weight percent on an oxide basis, of 40-57% SiO<sub>2</sub>, 2.0-11% Al<sub>2</sub>O<sub>3</sub>, 1-16% GaO, 8-21.5% SrO, 14-31.5% BaO, 0-3% MgO, and 0-4% B<sub>2</sub>O<sub>3</sub>, the glass having a temperature not over 1450°C at a viscosity of 20 MPa · s (200 poises), a CTE of 60-90x10<sup>-7</sup>/°C and a strain point over 630°C.



**FIG. 1**

EP 0 729 922 A1

FP03-0179 -0060-12
03.10.28
SEARCH REPORT

## Description

FIELD OF THE INVENTION

5 A plasma display device having an aluminosilicate glass panel exhibiting physical and chemical properties necessary for such devices.

BACKGROUND OF THE INVENTION

10 There is a rapidly growing interest in flat panel display devices. Thus far, commercial activity has centered on small units such as used in laptop computers. For this purpose, the liquid crystal display (LCD) device has been the dominant center of interest.

Increasing attention is being given to larger units that may be used in information and entertainment applications. LCDs tend to require critical accuracy in construction and, therefore, do not readily lend themselves to large size  
15 screens. Accordingly, as interest shifts to larger size units, attention is being directed to alternative types of display devices.

One such alternative is a plasma display device. In its simplest form, a plasma display device embodies two insulating glass substrates maintained in opposed, spaced relationship. One substrate has anode electrodes formed on its interface. The other substrate has cathode electrodes formed on its interface. Barriers are formed between the elec-  
20 trodes in each set to prevent cross talk.

The substrates are maintained in a fixed relationship facing each other. The electrodes are so arranged, and the substrates so aligned, that electrode segments face one another and correspond to a picture element. A rare gas, such as neon, argon, or helium is enclosed around and between the electrodes. When a voltage, which may be up to 100 V, is applied between the electrode sets, the gas undergoes a glow discharge. This is based on a principle commonly  
25 known as the neon glow discharge principle.

The light generated by this discharge is used to form the display. The electrodes may contain materials that generate the primary red, green and blue colors under influence of the discharge. In another form, fluorescent phosphors are coated on the electrodes and are affected by the discharge to produce the desired colors.

The electrodes may be applied in stripes that are perpendicular to one another. They may be printed, or may be  
30 formed by a photolithographic process. Alternatively, dots may be formed by printing, or by applying a film and processing the film by photolithographic techniques well known in that art.

Heretofore, the insulating substrates employed in emissive display devices have been sheets of soda lime glass. Soda lime glasses have been used because they provide a high coefficient of thermal expansion (CTE). A high CTE on the order of  $60-90 \times 10^{-7}/^{\circ}\text{C}$  ( $0-300^{\circ}\text{C}$ ) is required to match the expansion of glass frits used in producing electrodes and  
35 barriers in a display device.

For example, the electrodes and barriers may be applied as a paste and dried. The paste will contain a conductive component, such as a metal powder, a low melting point glass frit, and organics as a solvent and as a binder. The dried paste is fired to burn out any residual organic and to soften the glass frit to adhere to the substrate.

While soda lime glasses have the necessary high CTEs, they also have low strain points and low resistivities. Consequently, a soda lime substrate may shrink and/or undergo distortion during thermal processing. This processing  
40 includes firing the electrodes and/or sealing the substrates together. The high soda content also leads to sodium ion migration which degrades the display electronics (e.g. electrodes).

It would, therefore, be desirable to provide a glass substrate having a CTE in the  $60-90 \times 10^{-7}/^{\circ}\text{C}$  range and a strain point greater than  $630^{\circ}\text{C}$ . At the same time, it would be desirable to limit the use of alkalis in the substrate compositions since these ions may limit the lifetime of a display as well as lower the glass strain point. Most preferably, it would be  
45 desirable for the glass to be capable of being manufactured using the float process. It is a purpose of the present invention to provide glasses that meet these various conditions.

SUMMARY OF THE INVENTION

50 An aspect of the present invention is an aluminosilicate glass having a limited alkali metal ion concentration, whose composition comprises, as calculated in weight percent on an oxide basis, 40-57%  $\text{SiO}_2$ , 2.0-12 %  $\text{Al}_2\text{O}_3$ , 1-16%  $\text{CaO}$ , 8-21.5%  $\text{SrO}$ , 14-31 %  $\text{BaO}$ , 0-3%  $\text{MgO}$  and 0-4%  $\text{B}_2\text{O}_3$ , the glass having a temperature not over  $1500^{\circ}\text{C}$ , more preferably not over  $1450^{\circ}\text{C}$ , at a viscosity of  $20 \text{ MPa} \cdot \text{s}$  (200 poise), a CTE of  $60-90 \times 10^{-7}/^{\circ}\text{C}$  and a strain point over  $630^{\circ}\text{C}$ .  
55 These glasses preferably employ a total  $\text{R}_2\text{O}$  content (where  $\text{R} = \text{Li}, \text{Na}, \text{or K}$ ) which is less than about 5 weight percent.

A further aspect of the invention is a display panel having an aluminosilicate glass substrate that is free of alkali metal ions, that has a composition consisting essentially of, as calculated in weight percent on an oxide basis, 40-57%  $\text{SiO}_2$ , 2.0-11%  $\text{Al}_2\text{O}_3$ , 1-16%  $\text{CaO}$ , 8-21.5%  $\text{SrO}$ , 14-31.5%  $\text{BaO}$ , 0-3%  $\text{MgO}$  and 0-4%  $\text{B}_2\text{O}_3$ , the glass having a temperature not over  $1450^{\circ}\text{C}$  at a viscosity of  $20 \text{ MPa} \cdot \text{s}$  (200 poise), a CTE of  $60-90 \times 10^{-7}/^{\circ}\text{C}$  and a strain point over  $630^{\circ}\text{C}$ .

DESCRIPTION OF THE DRAWINGS

FIGURE 1 is a perspective, open view illustrating the basic structure of a plasma display panel, and  
FIGURE 2 is a fragmented, side elevation view in cross section of an element in FIGURE 1.

DESCRIPTION OF THE INVENTION

FIGURE 1 in the accompanying drawing is a perspective, open view illustrating the basic structure of a plasma display panel generally designated 10. Panel 10 comprises a front substrate 11 and a back substrate 12. Anode electrodes 13 are formed on the inner surface of front substrate 11, and cathode electrodes 14 are formed on the inner face of back substrate 12. Ribs 15 serve to space substrates 11 and 12, and also to act as isolating barriers. It will be appreciated that isolating barriers may take other known forms as well. In such case, substrates 11 and 12 may be spaced apart by a peripheral sealing strip rather than ribs 15.

FIGURE 2 is a fragmented, side elevation view in cross section. It shows, in greatly enlarged form, an element or cell 20 within display panel 10. Substrates 11 and 12 are joined together to form a hermetically enclosed panel. Then the space intermediate the substrates is evacuated and filled with a rare gas in known manner.

The present invention is primarily concerned with improved glasses that are adapted to production of insulating substrates, such as 11 and 12. In particular, the glasses meet the various property requirements of such substrates. In addition, many of the glasses of the present invention are also capable of being formed using the float glass manufacturing process. To provide flexibility in the melting process, and particularly to enable the production of the glasses via the float process, it is desirable that the glass have several characteristics related to melting properties. For example, a melting temperature (i.e., temperature at which the viscosity is about 20 MPa · s (200 poises)) less than or equal to about 1500°C, and more preferably less than or equal to about 1450°C, is desirable in some float processes. More importantly, in order to manufacture the glass via the float glass manufacturing process, it is desirable for the glass to exhibit a viscosity at the liquidus temperature which is greater than 50 MPa · s (500 poises), more preferably greater than 100 MPa · s (1000 poises), and most preferably greater than about 250 MPa · s (2500 poises).

The preferred glasses in accordance with the present invention have a CTE in the range of  $60-90 \times 10^{-7}/^{\circ}\text{C}$ , more preferably  $65-85 \times 10^{-7}/^{\circ}\text{C}$ , and most preferably  $79-85 \times 10^{-7}/^{\circ}\text{C}$ . The desire for such a CTE is primarily driven by the desire to match the CTEs of glass frits used in electrodes and sealing operations. The glasses of the present invention preferably have a strain point greater than 630°C, more preferably greater than 645°C, and most preferably greater than about 655°C. A high strain point is desired to help prevent panel distortion due to compaction/shrinkage during subsequent thermal processing. Such processing includes firing of electrodes, sealing of panels and application of coatings. In the most preferred embodiments, the glasses exhibit a combination of desirable CTE's and strain point. For example, the most preferred glasses exhibit a CTE in between  $79-85 \times 10^{-7}/^{\circ}\text{C}$ , in combination with a strain point greater than 645°C, and most preferably greater than about 655°C.

In one embodiment of the present invention, alkali metal oxides are essentially avoided other than at levels of under 1.0 weight percent. Glasses having such low alkali metal levels are particularly useful as substrates in direct current (D/C) plasma display panels. However, alkali metals can be utilized in glasses for other embodiments, e.g. plasma displays which are powered by alternating current. On the other hand, it is still desirable that the total alkali level be below about 5 weight percent.

The glass compositions employed herein utilize substantial amounts of the alkaline earth metal oxides BaO, SrO and CaO. BaO, which is the largest of these cations, has the greatest effect in increasing the CTE value, followed by SrO, CaO and MgO, in that order. Up to 3% MgO may be tolerated, but is generally avoided because it may cause phase separation in the glass and because it lowers the CTE excessively. The glasses of the present invention therefore employ 14-31.5% BaO, 8-21.5% SrO, 1-16% CaO and 0-3% MgO, all in weight percent. Preferably, these glasses also employ a total amount of these ingredients (BaO, SrO, MgO, and CaO) in an amount between about 33 and 55 weight percent, more preferably about 35 to 47 weight percent.

Chemical durability of the glasses herein has been determined as weight loss in  $\text{mg}/\text{cm}^2$  that occurs during immersion of a glass sample in 5% HCl solution at 95°C for 20 minutes. Chemical durability generally improves as the ratio of glass formers plus intermediates to glass modifiers increases. However, it is frequently necessary to effect a compromise between chemical durability and meltability. It is desirable that glasses for present purposes have a weight loss less than about  $5.0 \text{ mg}/\text{cm}^2$ , more preferably less than  $2.0 \text{ mg}/\text{cm}^2$  and most preferably less than  $1.0 \text{ mg}/\text{cm}^2$  after exposure to the 5% HCl solution as described above. Glasses having compositions within the following oxide ranges in weight percent are generally characterized by wt. loss values less than  $5.0 \text{ mg}/\text{cm}^2$ :

SiO <sub>2</sub>	44-57
Al <sub>2</sub> O <sub>3</sub>	2-12
CaO	1-15
SrO	9-19
BaO	16-29
MgO	0-3
B <sub>2</sub> O <sub>3</sub>	0-1

The present glasses employ 40-57% by weight SiO<sub>2</sub> as the primary glass former. Increasing SiO<sub>2</sub> content generally improves durability, but raises the melting point. The glasses also comprise 2.0-11% Al<sub>2</sub>O<sub>3</sub>. As the Al<sub>2</sub>O<sub>3</sub> content increases, glass durability increases, but CTE decreases and the melting point increases. Boric oxide (B<sub>2</sub>O<sub>3</sub>) decreases melting temperature, but is generally detrimental to durability, strain point, and CTE. Therefore, B<sub>2</sub>O<sub>3</sub> is limited to 4% more preferably is less than 2% and is most preferably omitted.

The invention is further illustrated by the following examples, which are meant to be illustrative, and not in any way limiting, to the claimed invention. TABLE I sets forth exemplary glass compositions in weight percent, as calculated on an oxide basis from the glass batches. These example glasses were prepared by melting 1000 gram batches of each glass composition at a temperature and time to result in a relatively homogeneous glass composition, e.g. at a temperature of about 1450-1550°C for a period of about 6 to 12 hours. Also set forth are relevant glass properties for each glass composition. These include linear coefficients of thermal expansion over the temperature range of 0-300°C (CTE $\times 10^{-7}/^{\circ}\text{C}$ ); strain point in °C; viscosity at the liquidus (Liq. Visc. MPa  $\cdot$  s/poises); temperature at viscosity of 2300 MPa  $\cdot$  s in °C (Visc. Temp.); melting temperature (Melt. Temp. at 20 MPa  $\cdot$  s); weight loss after exposure to 5% HCl solution at 95°C for 20 minutes (mg/cm<sup>2</sup>).

TABLE I

5		1	2	3	4	5	6	7
	SiO <sub>2</sub>	47.08	48.03	48.72	52.82	47.43	46.57	46.91
	Al <sub>2</sub> O <sub>3</sub>	7.53	5.65	4.3	8.45	8.37	8.22	8.28
10	CaO	6.08	6.21	6.3	8.53	4.6	4.51	4.55
	SrO	12.24	12.49	12.67	13.73	12.33	13.92	12.2
	BaO	27.07	27.62	28.01	16.47	27.27	26.78	26.97
15	MgO	--	--	--	--	--	--	1.09
	B <sub>2</sub> O <sub>3</sub>	--	--	--	--	--	--	--
20								
	CTE x10 <sup>-7</sup> /°C	73.7	75.8	76.9	66.5	72.1	72.6	72.8
25	Strain Point (°C)	673	667	660	680	676	675	676
	Liq. Visc. (MPa·s/ poises)	2,800 28,000	1,900 19,000	1,500 15,000	1,600 16,000	2,000 20,000	950 9,500	450 4,500
30								
	Visc. Temp. (°C at 2300 MPa·s)	1088	1075	1068	1120	1105	1096	1100
35								
	Melt Temp. (°C at 20 MPa·s)	1373	1357	1337	1413	1391	1375	1376
40								
	Wt. Loss	0.369	0.241	0.36	0.064	0.23	0.39	0.44

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EP 0 729 922 A1

		8	9	10	11	12	13	14
5	SiO <sub>2</sub>	43.98	47.28	56.91	48.79	53.73	47.07	46.1
	Al <sub>2</sub> O <sub>3</sub>	7.03	7.56	5.02	5.74	8.59	3.46	2.71
	CaO	4.26	4.58	5.52	4.73	6.95	13.69	14.85
10	SrO	18.4	12.29	14.8	12.69	13.97	9.85	11.92
	BaO	25.29	27.18	17.75	28.05	16.75	24.98	24.45
	MgO	1.02	1.1	--	--	--	--	--
15	B <sub>2</sub> O <sub>3</sub>	--	--	--	--	--	0.95	--
20	CTE x10 <sup>-7</sup> /°C	78.3	72.3	64.7	73.5	64.6	82.0	85.0
	Strain Point (°C)	676	673	673	662	682	666	668
25	Liq. Visc. (MPa·s/ poises)	120 1,200	2,200 22,000	150 1,500	4,500 45,000	8,000 80,000	24 240	130 1,300
30	Visc. Temp. (°C at 2300 MPa·s)	1068	1093	1137	1087	1134	1023	1023
35	Melt Temp. (°C at 20 MPa·s)	1318	1369	1449	1365	1451	1243	1236
40	Wt. Loss	4.1	0.27	0.017	0.137	0.04	3.8	4.53

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EP 0 729 922 A1

		15	16	17	18	19
5	SiO <sub>2</sub>	43.8	45.1	47.0	48.4	44.8
	Al <sub>2</sub> O <sub>3</sub>	6.2	6.4	6.3	6.5	8.4
	B <sub>2</sub> O <sub>3</sub>	—	—	—	—	—
10	Na <sub>2</sub> O	0.2	0.6	0.2	0.7	0.2
	K <sub>2</sub> O	—	1.0	—	0.9	—
	CaO	8.9	8.4	8.3	7.8	8.4
15	SrO	16.5	15.5	15.4	14.4	15.4
	BaO	24.4	23.0	22.8	21.3	22.8
20	CTE (x10 <sup>-7</sup> /° C)	84.5	85.2	82.0	81.9	78.7
	Strain Pt (° C)	673	654	675	646	678
25	Liq. Vis. (MPa·s/ poises)			350 3500		
	Liquidus (° C)	1195-	1170-	1150-	1145-	1150-
30	Temperature	1205	1200	1155	1170	1160
	Soft. Pt (° C)	882	869	882	873	894
	Anneal Pt (° C)	718	700	718	693	723
35						
40						
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		20	21	22	23	24
5	SiO <sub>2</sub>	48.1	49.0	47.3	47.5	50.6
	Al <sub>2</sub> O <sub>3</sub>	9.0	9.2	8.8	8.8	11.8
	B <sub>2</sub> O <sub>3</sub>	—	—	1.0	1.5	—
10	Na <sub>2</sub> O	0.7	0.7	0.7	0.7	1.4
	K <sub>2</sub> O	1.0	1.1	1.0	1.0	2.2
15	CaO	11.4	13.3	8.0	7.3	10.3
	SrO	12.0	10.8	13.4	13.4	9.6
	BaO	17.8	15.9	19.8	19.8	14.1
20	CTE (x10 <sup>-7</sup> /° C)	79.5	78.7	77.0	77.0	74.3
25	Strain Pt (° C)	661	664	645	637	645
	Liq. Visc.					
	Liquidus(° C) Temperature	1200- 1205	1220- 1240	1100- 1120	1190	1125- 1100
30	Soft. Pt (° C)	880	885	866	879	859
	Anneal Pt (° C)	708	710	692	684	694

A glass having a composition and properties as shown in Example 17 is currently regarded as representing the best mode of the invention, that is, as providing the best combination of properties for the purposes of the invention at this time.

#### Claims

1. A glass exhibiting a linear coefficient of thermal expansion (CTE) over the temperature range 0-300°C between 60-90x10<sup>-7</sup>/°C and a strain point over 630°C, said glass comprising an aluminosilicate having a composition as calculated in weight percent on an oxide basis, of 40-57% SiO<sub>2</sub>, 2.0-12% Al<sub>2</sub>O<sub>3</sub>, 1-16% CaO, 8-21.5% SrO, 14-31.5% BaO, 0-3% MgO, and 0-4% B<sub>2</sub>O<sub>3</sub>.
2. A glass in accordance with claim 1 which is manufactured by the float process, said glass having a liquidus viscosity greater than about 50 MPa · s (500 poises), and/or a CTE of 79-85x10<sup>-7</sup>/°C, and a strain point over 645°C.
3. A glass according to claim 1 or 2 wherein the B<sub>2</sub>O<sub>3</sub> content of said glass is less than about 1.5 weight percent, and the alkali metal content is less than about 5 percent by weight.
4. A glass according to claim 3, wherein the MgO content of said glass is less than about 1.5 weight percent.
5. A glass according to claim 2, wherein said glass has a strain point over 655°C.



**EP 0 729 922 A1**

6. A glass according to claim 1, 2, 3, 4 or 5, wherein said glass consists essentially of a composition, as calculated in weight percent on an oxide basis, of 40-57% SiO<sub>2</sub>, 2.0-12% Al<sub>2</sub>O<sub>3</sub>, 1-16% CaO, 8-21.5% SrO, 14-31.5% BaO, 0-3% MgO, and 0-4% B<sub>2</sub>O<sub>3</sub>, or 41-52% SiO<sub>2</sub>, 2-12% Al<sub>2</sub>O<sub>3</sub>, 4-16% GaO, 8-19% SrO, 16-29% BaO, 0-3% MgO, and 0-4% B<sub>2</sub>O<sub>3</sub>.

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7. A glass according to any one of claims 1-6, as a flat transparent substrate for a flat panel display device.

8. A substrate according to claim 4 and 7, wherein said glass has less than about 1.0 percent alkali metal content.

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9. The substrate of claim 7 or 8 in a direct current powered emissive flat panel display.

10. A substrate according to claim 7, 8 or 9, wherein the glass has a weight loss less than 5.0 mg/cm<sup>2</sup> when immersed in 5% HCl at 95°C for one hour

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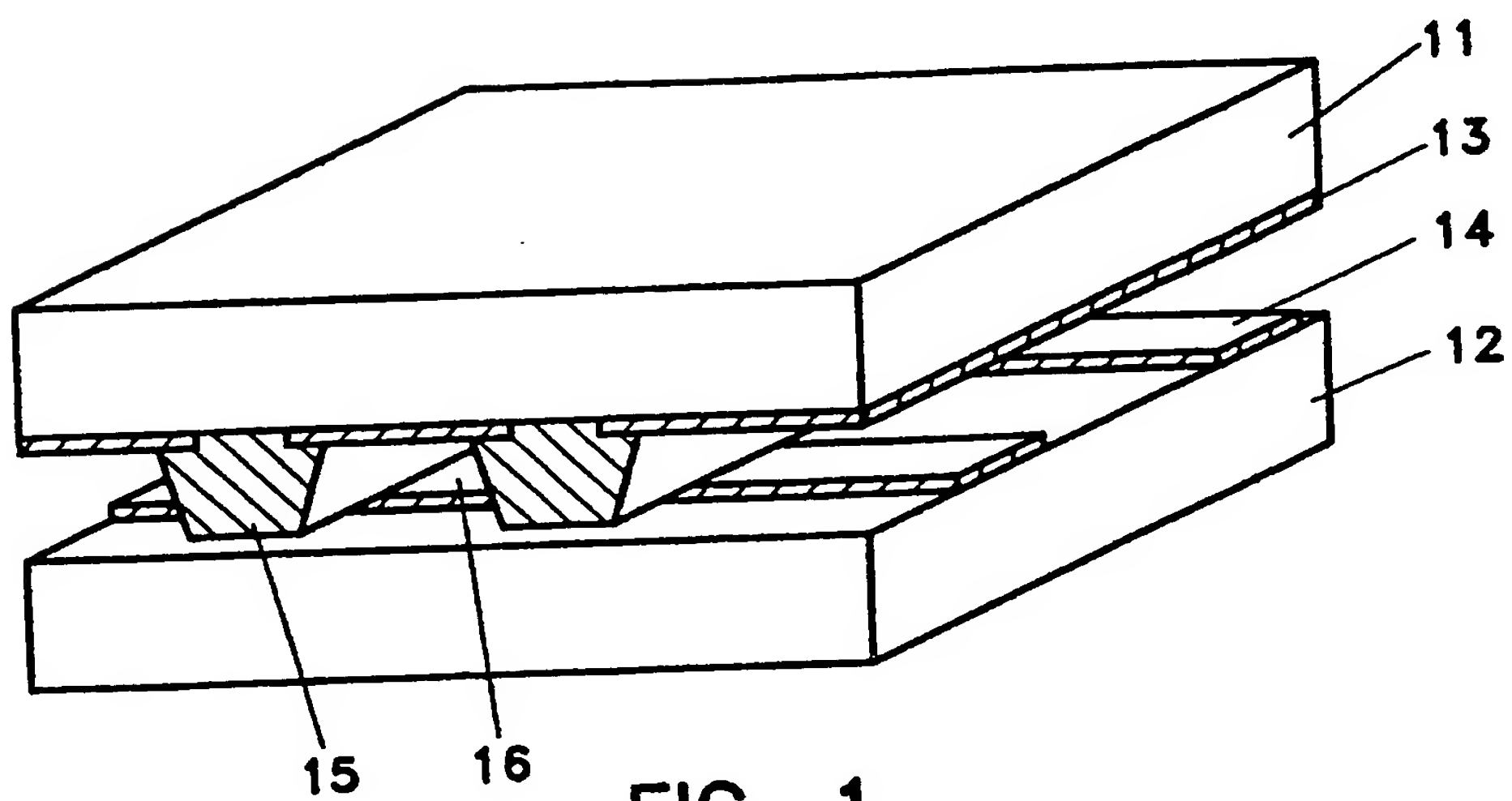


FIG. 1

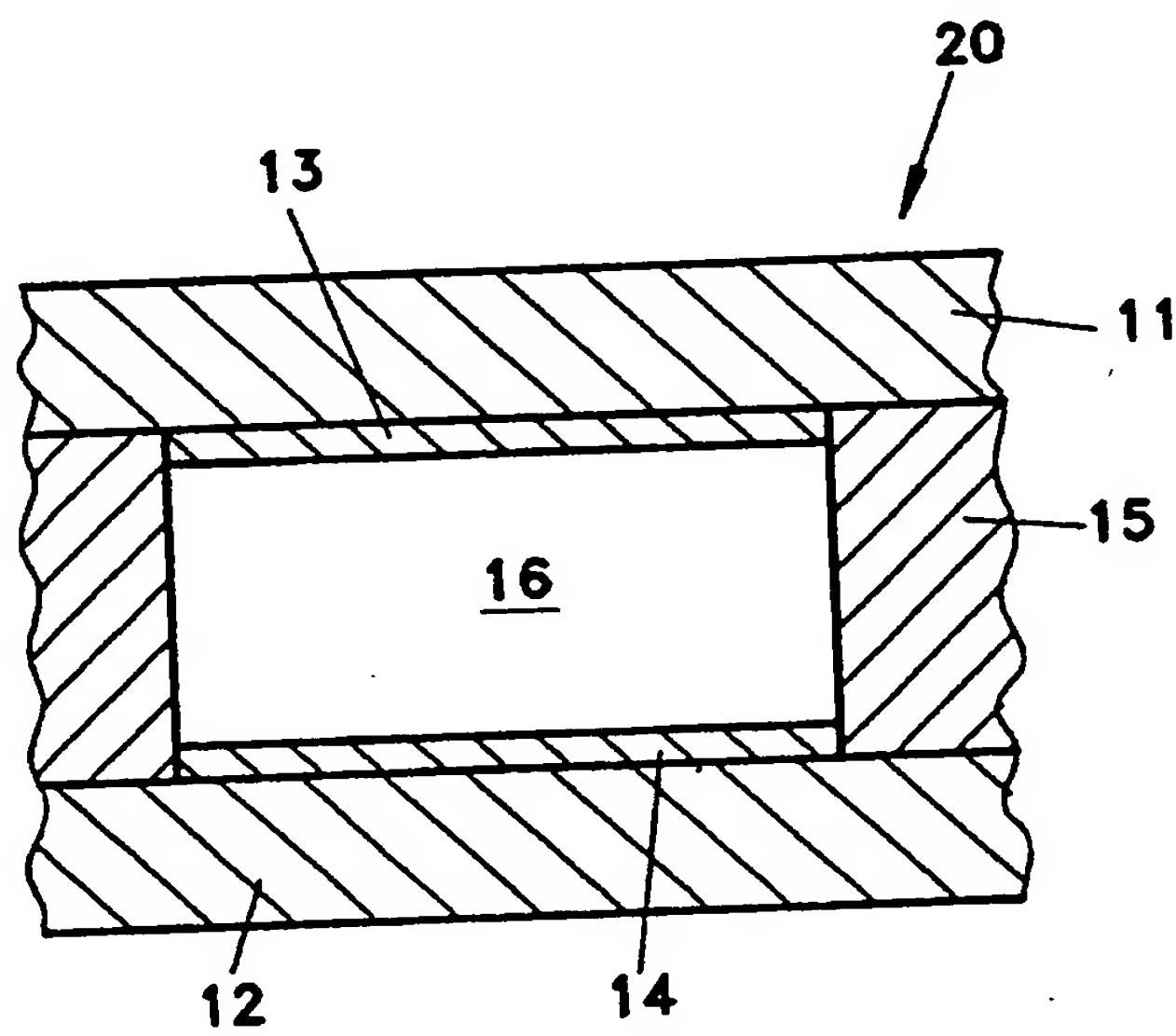


FIG. 2



European Patent  
Office

# EUROPEAN SEARCH REPORT

Application Number  
EP 96 10 2722

DOCUMENTS CONSIDERED TO BE RELEVANT			
Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	CLASSIFICATION OF THE APPLICATION (Int.Cl.6)
A	PATENT ABSTRACTS OF JAPAN vol. 016, no. 479 (C-0992), 6 October 1992 & JP-A-04 175242 (ASAHI GLASS CO LTD), 23 June 1992, * abstract *	1-10	C03C3/087 C03C3/091
A	--- US-A-5 326 730 (DUMBAUGH, JR.) 5 July 1994 * column 3, line 49 - column 4, line 61; table I *	1-10	
A	--- DE-A-37 30 410 (NIPPON ELECTRIC GLASS CO., LTD.) 7 April 1988 * page 3, line 6 - line 53; table 1 *	1-10	
			TECHNICAL FIELDS SEARCHED (Int.Cl.6)
			C03C
The present search report has been drawn up for all claims			
Place of search		Date of completion of the search	Examiner
THE HAGUE		7 June 1996	Van Bommel, L
<p><b>CATEGORY OF CITED DOCUMENTS</b></p> <p>X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document</p> <p>T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons</p> <p>&amp; : member of the same patent family, corresponding document</p>			

EPO FORM 1503 01/92 (P04C01)

# INTERNATIONAL SEARCH REPORT

International application No.

PCT/JP03/09051

## A. CLASSIFICATION OF SUBJECT MATTER

Int.Cl<sup>7</sup> G09F9/30, C04B35/16, H05B33/02, H05B33/14, C03C10/00, C03C10/06

According to International Patent Classification (IPC) or to both national classification and IPC

## B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

Int.Cl<sup>7</sup> G09F9/30-9/46, C04B35/16, H05B33/00-33/14, C04B35/00-35/22, C03C1/00-14/00

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Jitsuyo Shinan Koho	1922-1996	Jitsuyo Shinan Toroku Koho	1996-2003
Kokai Jitsuyo Shinan Koho	1971-2003	Toroku Jitsuyo Shinan Koho	1994-2003

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

## C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y 1.	US 5459109 A (Corning Inc.), 17 October, 1995 (17.10.95), Abstract; Claims & JP 10-506088 A & EP 782552 A1 & DE 69527796 E & WO 96/09259 A1 & KR 97706214 A & TW 308583 B	1-8
Y 2.	EP 729922 A1 (Corning Inc.), 04 September, 1996 (04.09.96), Abstract; Claims; page 4, lines 1 to 20 & JP 8-333133 A & US 5741746 A	1-8
Y 3.	JP 2000-72472 A (Nippon Sheet Glass Co., Ltd.), 07 March, 2000 (07.03.00), Abstract; Claims (Family: none)	1-8

☒ Further documents are listed in the continuation of Box C. ☐ See patent family annex.

\* Special categories of cited documents:

"A" document defining the general state of the art which is not considered to be of particular relevance  
 "E" earlier document but published on or after the international filing date  
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 "X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone  
 "Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art  
 "&" document member of the same patent family

Date of the actual completion of the international search  
10 October, 2003 (10.10.03)

Date of mailing of the international search report  
28 October, 2003 (28.10.03)

Name and mailing address of the ISA/  
Japanese Patent Office

Authorized officer

Facsimile No.

Telephone No.

## INTERNATIONAL SEARCH REPORT

International application No.

PCT/JP03/09051

## C (Continuation). DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y 4.	JP 11-135899 A (Kyocera Corp.), 21 May, 1999 (21.05.99), Abstract; Claims (Family: none)	1-8
Y 5.	JP 10-70214 A (Kyocera Corp.), 10 March, 1998 (10.03.98), Abstract; Par. Nos. [0006] to [0010], [0025] to [0026], [0039] to [0041] (Family: none)	1-8
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Y 7.	JP 63-307141 A (Asahi Glass Co., Ltd.), 14 December, 1988 (14.12.88), Claims; page 2, upper left column, line 12 to upper right column, line 13 (Family: none)	1-8
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Y 10.	JP 58-29880 A (Matsushita Electric Industrial Co., Ltd.), 22 February, 1983 (22.02.83), Claims; page 2, upper left column, line 13 to upper right column, line 13; Fig. 1 (Family: none)	6-8